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## Composition Effects on the Critical Current of MOCVD-processed Zr:GdYBCO Coated Conductors in an Applied Magnetic Field

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# Introduction

- $I_c(B)$  of YBCO coated conductors can be modified through
  - Changing composition (RE:Ba:Cu ratios)
  - Rare earth substitution (replacing Y with Gd, Sm)
  - Adding extrinsic defects (Zr, Sn & more being searched)
- There have been many studies on the composition effects for REBCO, but little has been done for Zr:REBCO
- It is crucial to improve the understanding on the interaction between the intrinsic and extrinsic pinning centers

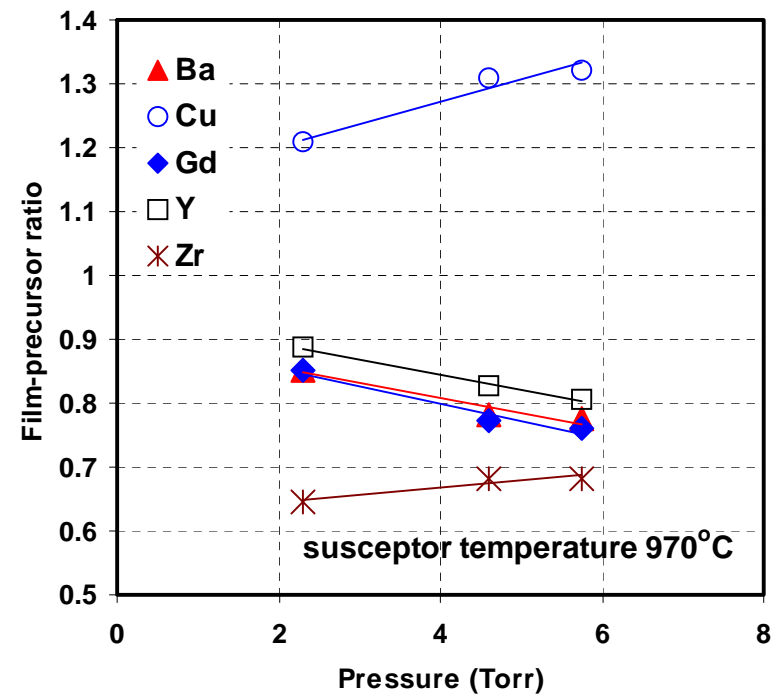
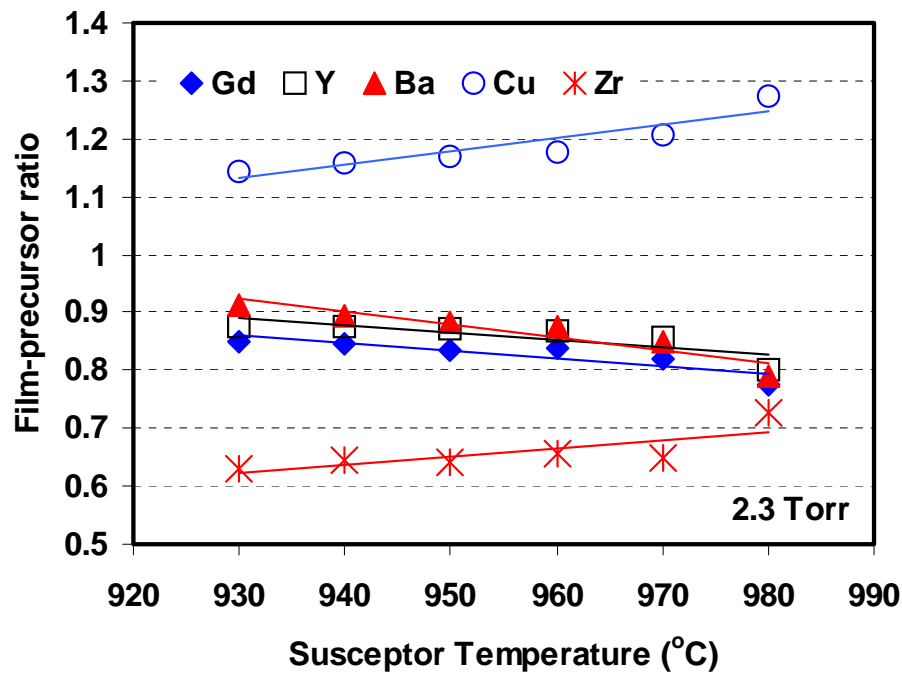
# Some of the advancements in understanding

- RE<sub>2</sub>O<sub>3</sub> precipitates
  - Lu, Li, Zhao (APL 1992): Y<sub>2</sub>O<sub>3</sub> nanodots are effective pinning for 0T – 7T
  - Wang (J. Appl. Phys. 2006): Y<sub>1.1</sub> for optimum J<sub>c</sub>(B)
  - Chen, Selva (APL 2009): RE<sub>2</sub>O<sub>3</sub> dots could be assembled to form layers, which were aligned with ab-plane and enhanced pinning in ab-plane
- BaZrO<sub>3</sub> columns
  - Driscoll (Nature Mat 2004): Zr extrinsic defects, strong pinning along c-axis
  - Goyal (Superc. S&T 2005): nano-columns comprised of BZO dots
  - Yamada (APL 2005): BZO nano-columns through the film thickness
  - Chen, Selva (APL 2009): interaction of bidirectionally aligned defects; explained J<sub>c</sub>(B)
- In the present work, we want to improve our understanding of the Interaction between RE<sub>2</sub>O<sub>3</sub> and BZO via investigating the composition effects:
  - Effects of Gd+Y content
  - Effects of Gd to Y ratio
  - Effects of Zr concentrations

## Composition Control in MOCVD for Zr:GdYBCO

- Our control is based on the functional relation of the film composition with the precursor composition, deposition temperature and pressure.
- Incorporate rate or film-precursor ratio depends on T and P

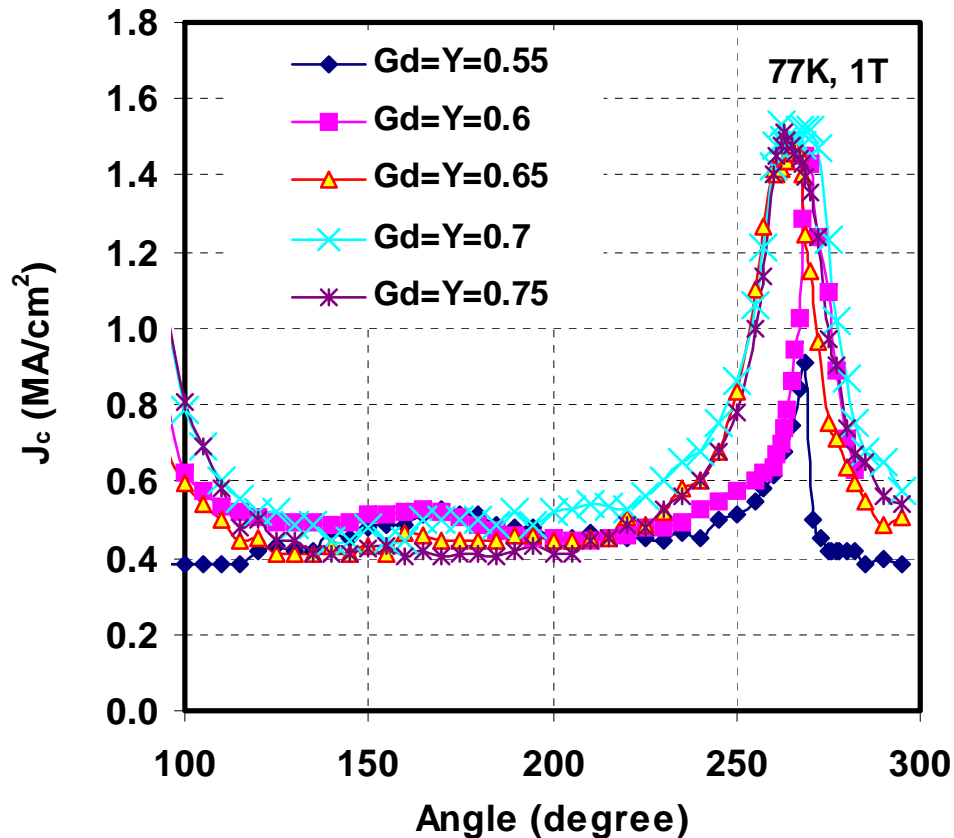
For example, precursor  $\text{Zr}_{0.065}\text{Gd}_{0.6}\text{Y}_{0.6}\text{Ba}_2\text{Cu}_{2.3}$  + Susceptor  $965^\circ\text{C}$  + Pressure 2.3Torr = film of  $\text{Zr}_{0.05}\text{Gd}_{0.6}\text{Y}_{0.6}\text{Ba}_2\text{Cu}_{3.3}\text{O}_7$



# Effects of Gd+Y content in GdYBCO

Rare-earth rich eliminate Ba-Cu-O phase; enhance pinning.

Aligned RE<sub>2</sub>O<sub>3</sub> dots enhance ab-peak.

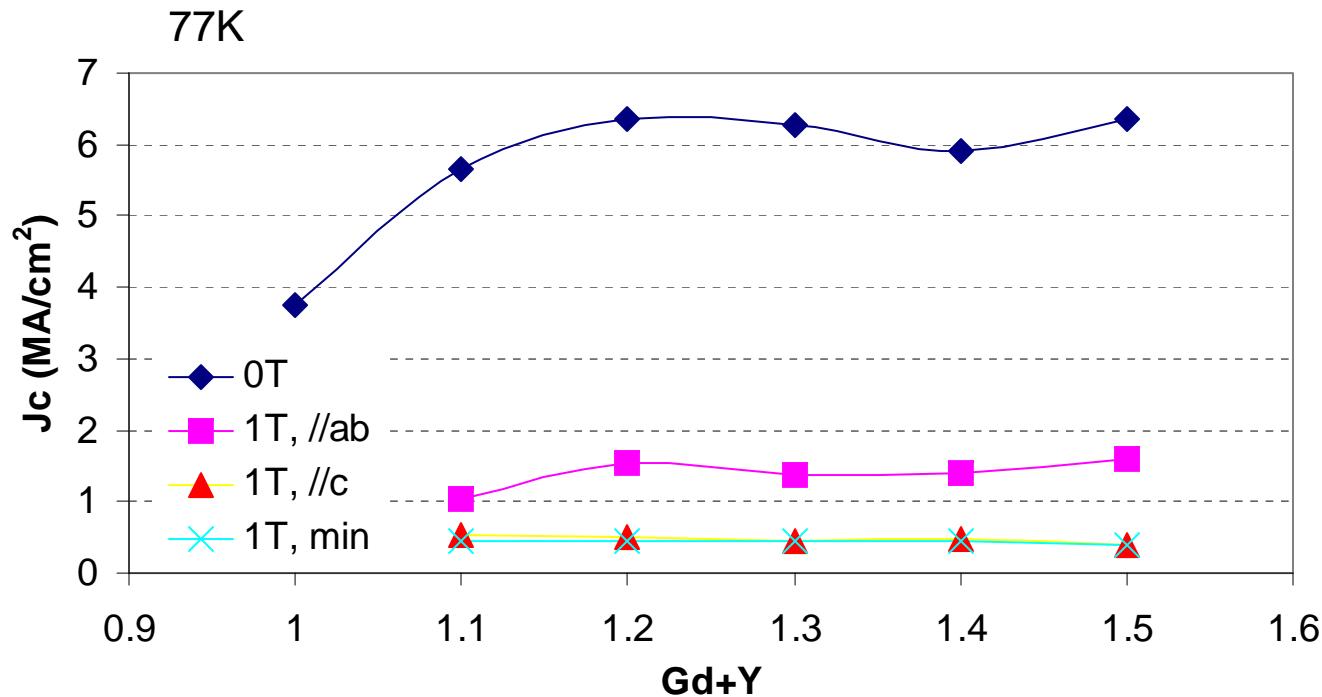


Varying Gd+Y in 0.5 μm thick GdYBCO film with Ba=2, Cu=3.3

**Gd+Y content dependence of  $J_c(sf)$ ,  $J_c(1T, //ab)$ ,  $J_c(1T, //c)$ ,  $J_{c\_min}(1T, any\ angle)$**

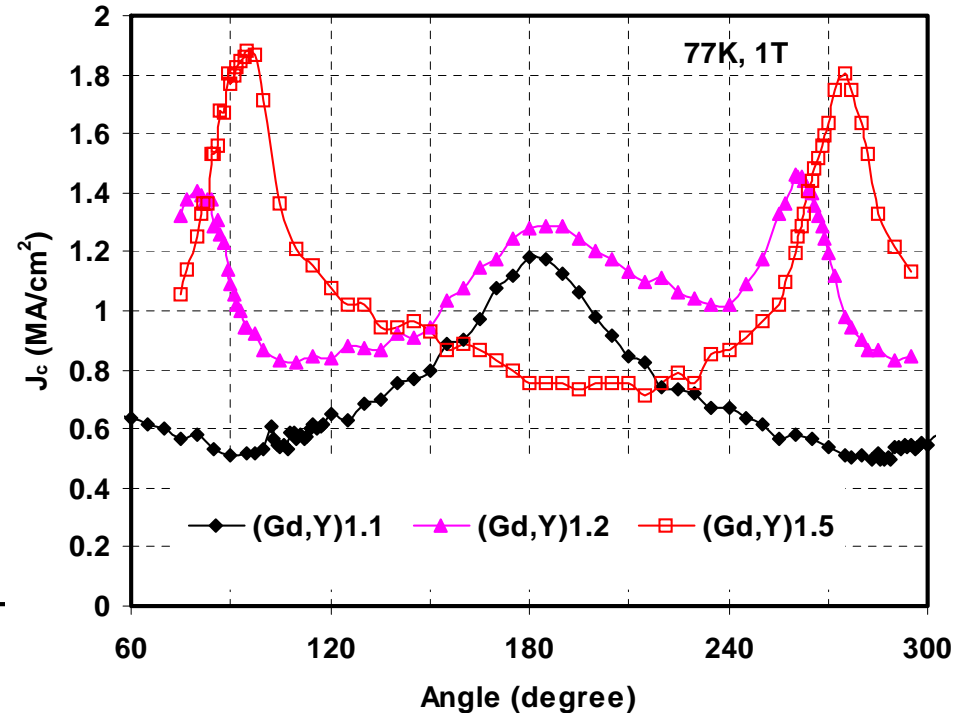
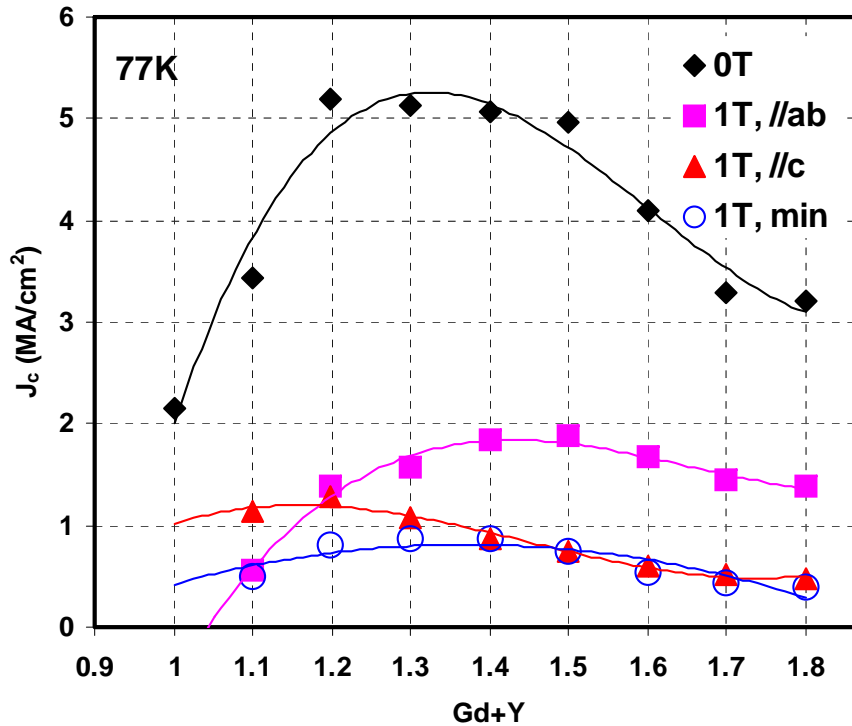
In a wide range of rare-earth content (RE = 1.2 – 1.5),  $J_c$  varied not much.

This finding is in contrast to that reported by Wang for PLD YBCO, where  $J_c$  dropped dramatically when Y content varying from 1.1 to 1.3.



**Gd+Y content dependence for in 0.5  $\mu\text{m}$  thick GdYBCO film with Ba=2, Cu=3.3**

# Effects of Gd+Y content in Zr:GdYBCO



RE content dependence for 0.5  $\mu\text{m}$  thick Zr:GdYBCO film with Ba=2, Cu=3.3, Zr=0.055

The c-peak was leveled off completely when RE>1.4.

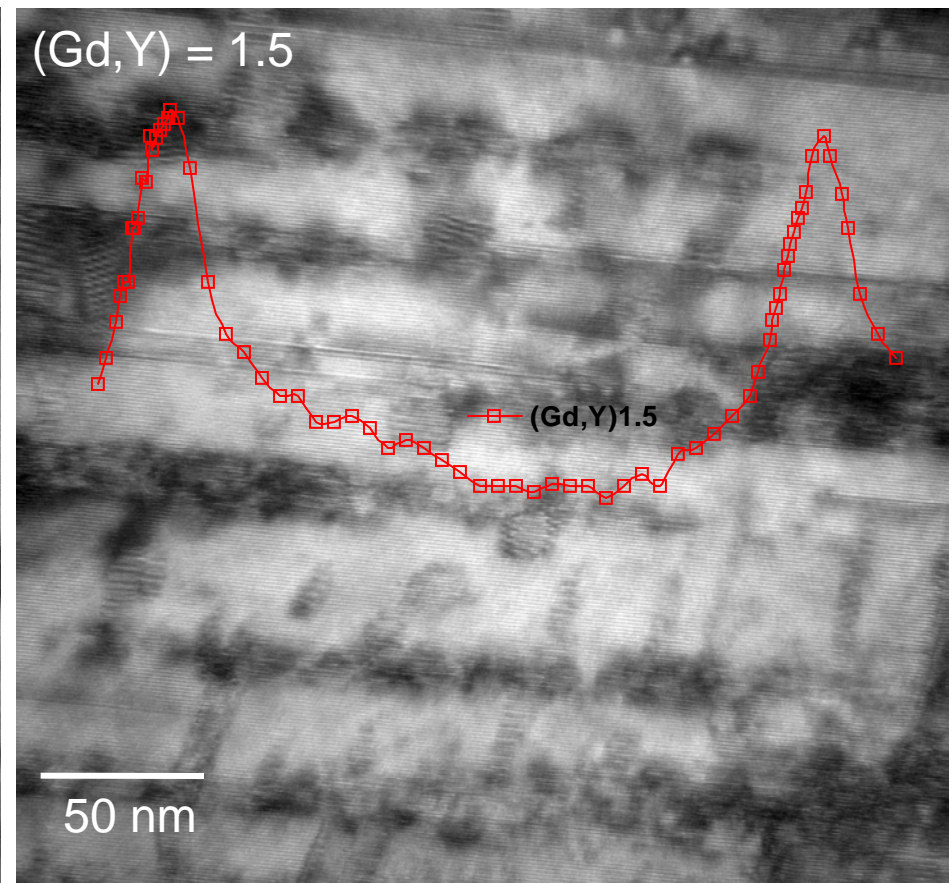
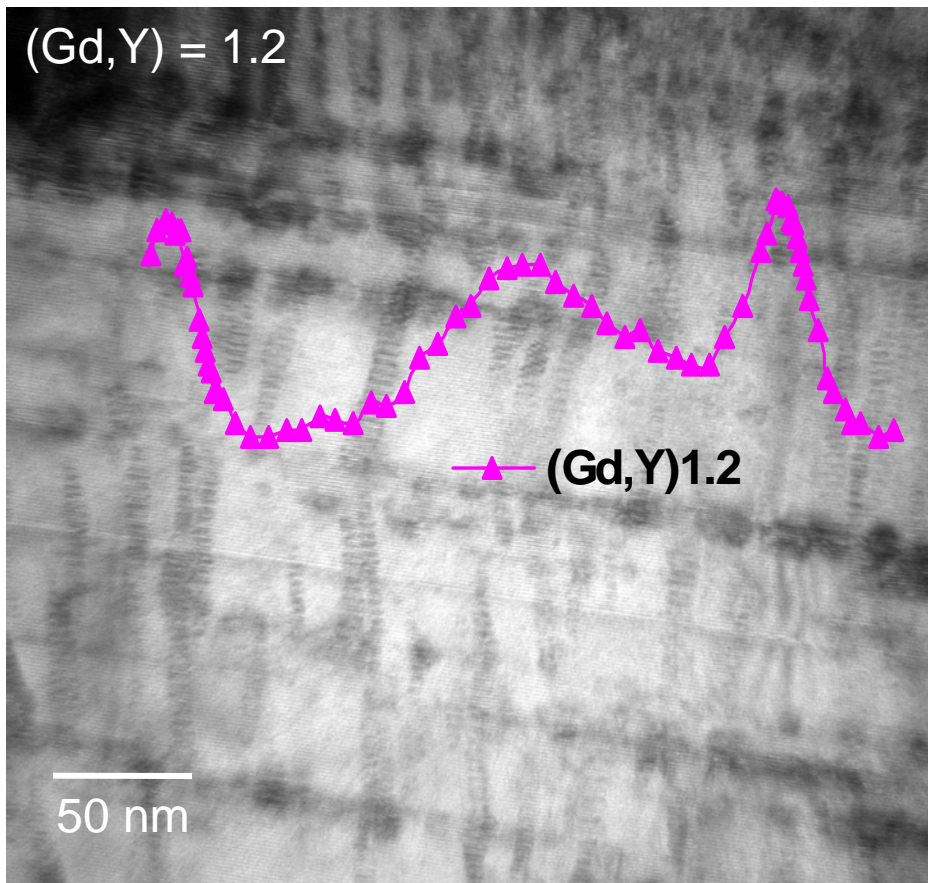
The a-b peak could also be leveled off under certain conditions.

However, the  $J_{c\_min}(B=1T)$  is still about 2 times of that without Zr.

# Interaction between vertically aligned BZO columns and horizontally aligned RE<sub>2</sub>O<sub>3</sub> precipitates

RE=1.2 content shows both columnar defects and layers comprised of nano-precipitates

RE=1.5 exhibits only short columnar segments; instead “thicker” precipitates are aligned with ab-planes

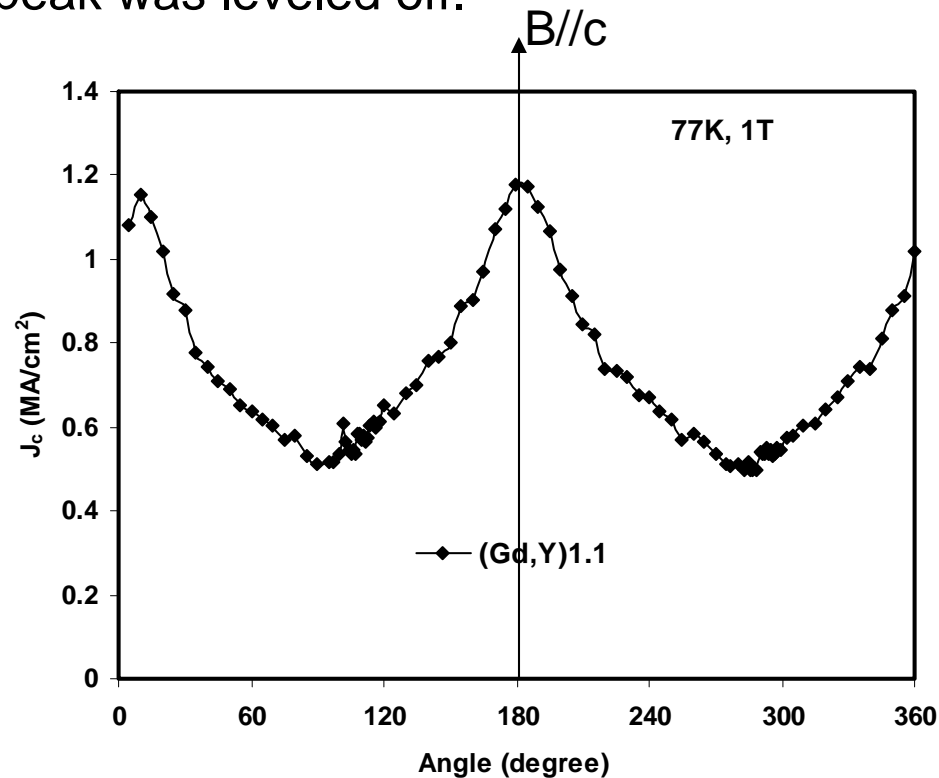


What happened to RE1.1? How was the ab-peak leveled off?

Lacking excess RE?

No. 10% is a large amount. In the case of without Zr addition, the ab-peak constantly stand there.

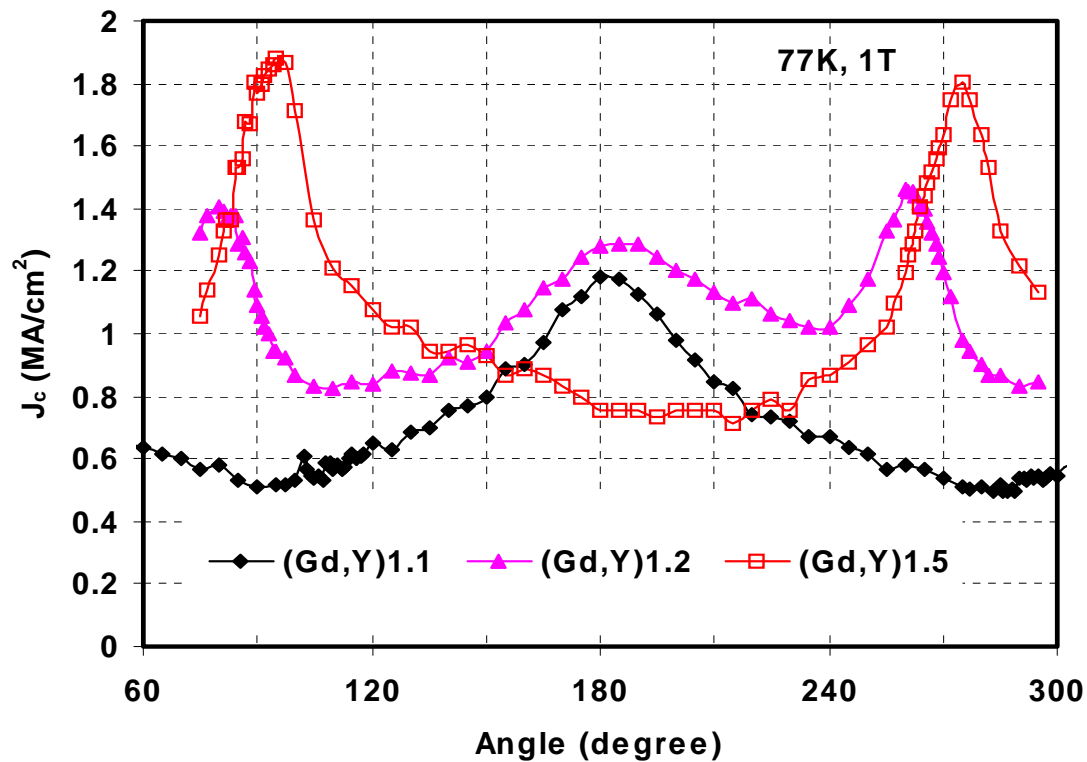
Under certain growth condition, the well formed BZO nanocolumns penetrating through the film thickness destroyed the horizontal alignment. Thus, the ab-peak was leveled off.



Can the superposition of the curves for low RE and high RE make a flat curve?

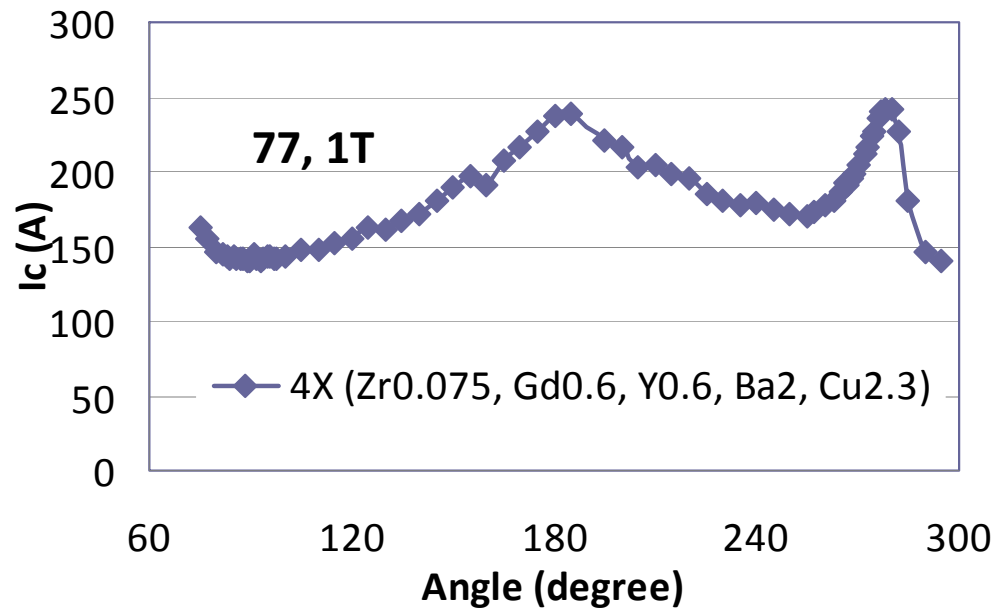
We can test it. However, it will not result in a higher  $J_{c\_min}$ .

We are more interested in seeing the superposition with RE1.2.



# In thicker Zr:GdYBCO films

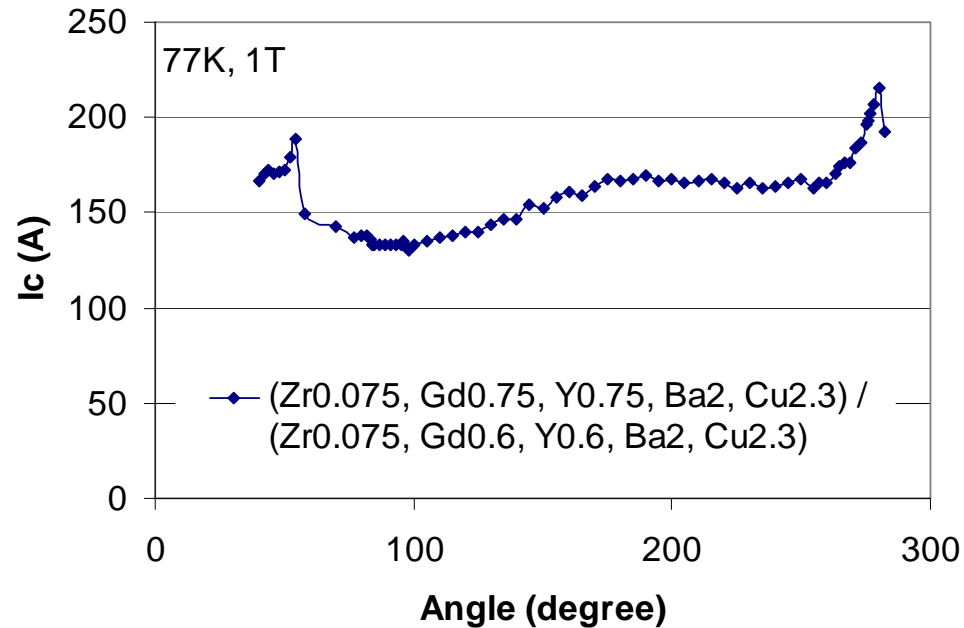
- 4 times-thick film of  $Zr_{0.055}Gd_{0.6}Y_{0.6}Ba_2Cu_{3.3}O_7$ , processed by 4 passes enhanced c-peak



B=	0	1T, //ab	1T, //c	1T
I <sub>c</sub> (A)	529	241	237	142

## In thicker Zr:GdYBCO films

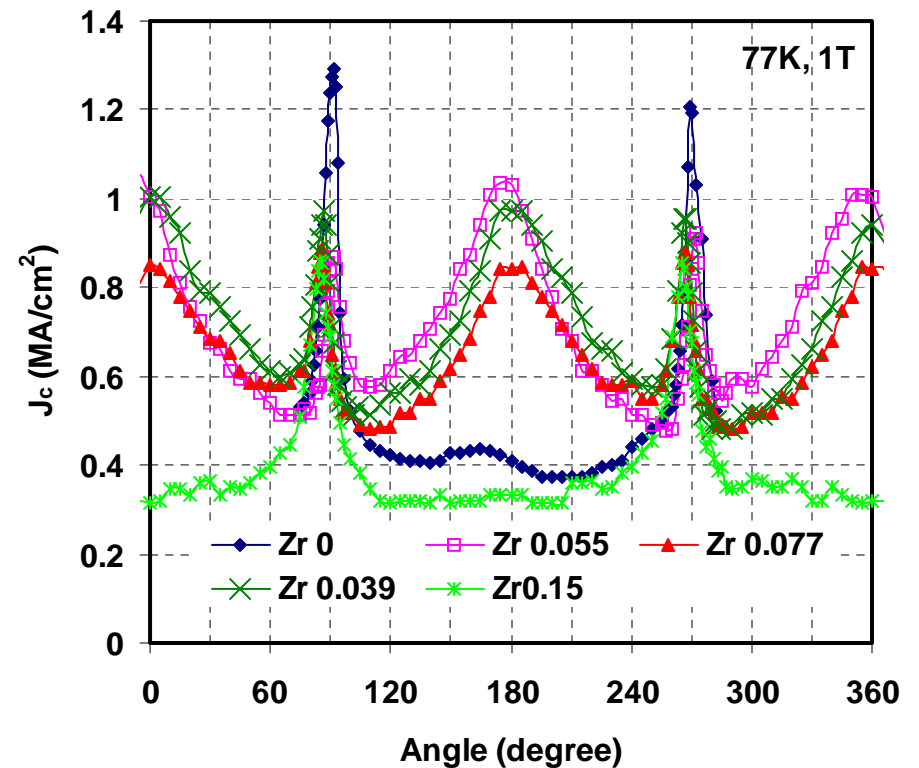
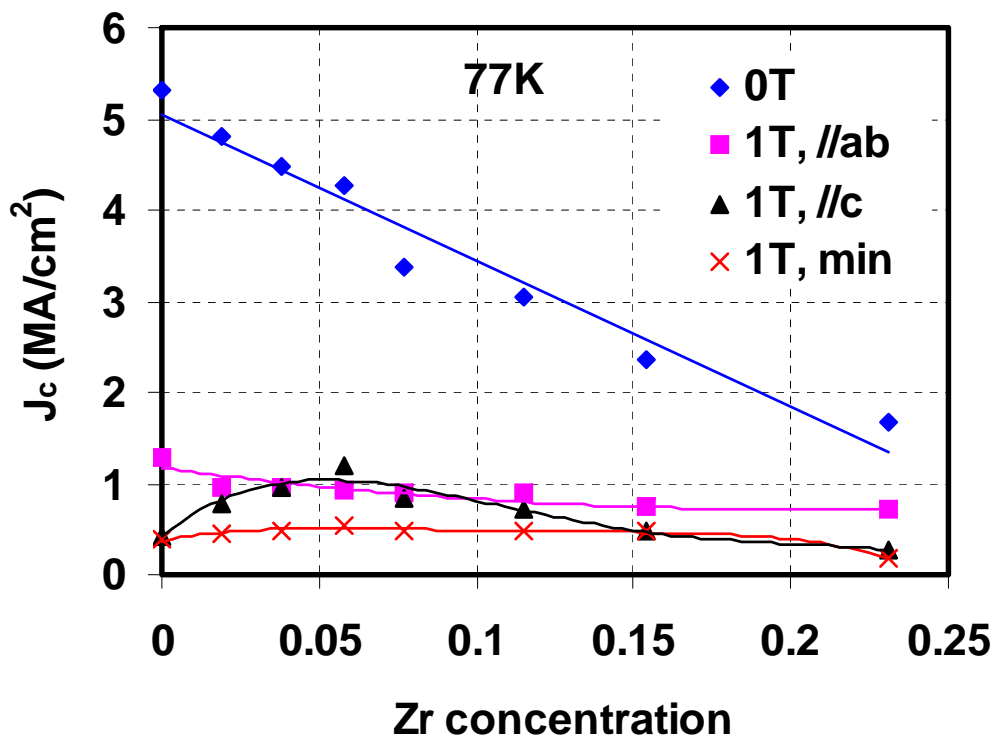
- Alternating composition from pass to pass
- $Zr_{0.06}Gd_{0.75}Y_{0.75}Ba_2Cu_{3.3}O_7 \setminus Zr_{0.06}Gd_{0.6}Y_{0.6}Ba_2Cu_{3.3}O_7$
- $Zr_{0.06}Gd_{0.75}Y_{0.75}Ba_2Cu_{3.3}O_7 \setminus Zr_{0.06}Gd_{0.6}Y_{0.6}Ba_2Cu_{3.3}O_7$
- The c-peak is still leveled off



B=	0	1T, //ab	1T, //c	1T
$I_c$ (A)	585	215	167	130

## Zr concentration effects in Zr:GdYBCO

For Zr-concentration in the range of 0.04 – 0.075, c-peak can be higher than or the same height as ab-peak.  $J_c(1T, \text{min})$  also takes its optimum value for Zr-concentration in this range,  $\sim 0.8 \text{ MA/cm}^2$ , which is double of the value for GdYBCO without Zr doping.



Zr-concentration dependence of  $J_c$  for in  $0.5 \mu\text{m}$  Zr:GdYBCO film with Gd=0.6, Y=0.6, Ba=2, Cu=3.3

# Conclusion

- The composition ratio RE:Ba:Cu is the key to effective flux pinning of Zr-inclusions in MOCVD fabricated REBCO films. Over-high RE could suppress or level off c-peak associated with the pinning from BZO nano-columns; Insufficient RE leads to low  $I_c$
- The removal of  $J_c$  peak at B//ab could be attributed to broken of horizontal alignment of  $RE_2O_3$  dots by the well-formed BZO columns
- RE content greater than 1.4 could level off the  $J_c$  peak at B//c
- For RE content ranging from 1.2 to 1.5,  $J_c(sf)$  and  $J_{c\_min}(77K, 1T)$  vary little
- The Zr-concentration in the range of 0.025 – 0.075 could be effective pinning centers
- Increasing Gd/Y ratio increased  $J_c(1T, 77K)$ . In self-field or low field, however, the optimized Gd/Y ratio is about 1



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